

# CoS<sub>2</sub> Yolk-Shell Spheres Coated with Carbon Thin Layers as High Active and Stable Electrocatalysts for Hydrogen Evolution Reaction

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**Abstract:** Though water electrolysis is effective in generating high-quality hydrogen gas, it requires effective electrocatalysts for hydrogen evolution reaction (HER). CoS<sub>2</sub> have been considered as a promising HER electrocatalyst because of its high catalytic activity. However, the key limitation for CoS<sub>2</sub> nanomaterial as HER electrocatalyst is its poor stability, which may be due to the structural breakdown of CoS<sub>2</sub> nanostructure or the evolution of S during H<sub>2</sub> evolution in acid media. Coating porous carbon thin layer for protection from structural breakdown and evolution of S is a good way to improve catalytic stability. In addition, coating carbon layer can change electronic structure of CoS<sub>2</sub> for the moderated hydrogen adsorption energy, leading to enhanced catalytic activity. Here, CoS<sub>2</sub> yolk-shell spheres coated with carbon thin layers exhibit superior catalytic performance for HER with low overpotential, small Tafel slope, and excellent stability.

**Key words:** CoS<sub>2</sub>@C; yolk-shell sphere; electrocatalyst; hydrogen evolution reaction (HER); high stability

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## 0 Introduction

Hydrogen as a sustainable, secure, clean and alternative energy source can ease the energy crisis and environment pollution faced in the present world<sup>[1-2]</sup>. Electrochemical water splitting is considered as a highly effective method to produce hydrogen<sup>[3]</sup>. However, it needs electrocatalysts to reduce overpotential due to low kinetics of water splitting. Pt is at present the most active catalyst for the hydrogen evolution reaction (HER)<sup>[4-5]</sup>. However, its scarcity on the earth and high cost greatly restrict the industrial production of hydrogen. Therefore, it is necessary and urgent to find cost-effective and earth-abundant catalysts with high HER catalytic activity and excellent stability to facilitate translation of H<sub>2</sub>O to hydrogen.

Transition metal dichalcogenides with gener-

alized formula of MX<sub>2</sub> (M refers to transition metal; X represents a chalcogen such as S and Se) have received increasing interests due to their low cost and high abundance<sup>[6-10]</sup>. Among them, MoS<sub>2</sub> has been considered as a promising candidate because of its high activity and stability<sup>[11-14]</sup>. However, the experimental and computational studies have concluded that the catalytic activity mainly arises from the active sites located along the edges of 2-D MoS<sub>2</sub> layers which are under-coordinated and thermodynamically unfavorable, and the basal planes are catalytically inert<sup>[15-17]</sup>. Although various methods have been adapted to expose more edge sites or enhance the intrinsic activity of the edge sites, the enhancement of HER electrocatalytic performance of MoS<sub>2</sub> still faces enormous challenges<sup>[18-20]</sup>. Another transition metal dichalcogenides CoS<sub>2</sub>, which was often utilized as electrode materials for supercapacitors,

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Li-ion batteries or electrocatalysts for oxygen reduction reaction (ORR)<sup>[21-26]</sup>, also has shown excellent HER electrocatalytic activity, even higher than MoS<sub>2</sub>. In addition, CoS<sub>2</sub>, which is an intrinsically conductive metal in contrast to the other MX<sub>2</sub> such as FeS<sub>2</sub><sup>[27]</sup> and NiS<sub>2</sub><sup>[28]</sup>, can improve the electron transportation from catalyst surface to the electrode, thus will reduce the overpotential needed to overcome the energy barriers and decrease the energy consumption. What's more, CoS<sub>2</sub> can exhibit high HER activity after conversion from thermodynamically favored semiconducting phase to a metastable metallic polymorph compared with other transition metal dichalcogenides, such as MoS<sub>2</sub> and WS<sub>2</sub><sup>[29-32]</sup>. However, the key limitation for CoS<sub>2</sub> as HER electrocatalyst is poor stability in acid media, which may be due to the structural breakdown of CoS<sub>2</sub> nanostructure or the evolution of S during H<sub>2</sub> evolution<sup>[24,33-37]</sup>. The poor stability of CoS<sub>2</sub> materials has seriously restricted their practical applications as high-performance HER electrocatalysts. Therefore, it is significant to develop CoS<sub>2</sub>-based electrocatalysts with excellent stability as well as high catalytic activity.

To improve the electrocatalytic activity and stability of CoS<sub>2</sub> for HER, carbon thin layer coating will be a promising method because of following advantages: (1) the carbon thin layer can well prevent CoS<sub>2</sub> from structural breakdown and the evolution of S; (2) the electronic interaction between CoS<sub>2</sub> and thin carbon layer will change the electronic structure of CoS<sub>2</sub>, which will be beneficial for the improvement of electrocatalytic activity; (3) The carbon layer will provide "super-highways" for electron transfer to promote HER

due to its high electrical conductivity. Based on the above considerations, we devote our attention to designing and synthesizing the novel CoS<sub>2</sub> yolk-shell (YS) spheres coated with carbon thin layers (CoS<sub>2</sub> YS@C spheres) as highly efficient HER electrocatalysts by a simple hydrothermal method. The CoS<sub>2</sub> YS@C spheres own hollow structure and high specific surface area, and they exhibit excellent catalytic activity with low onset potential of only about 20 mV, small Tafel slope of about 55 mV/dec, and small overpotential of about 90 mV at 10 mA/cm<sup>2</sup> in acidic solution. Especially, CoS<sub>2</sub> YS@C spheres also exhibit excellent stability at 10 mA/cm<sup>2</sup> for 10 h. This work provides a new revenue for the development of CoS<sub>2</sub>-based electrocatalysts with high catalytic activity and excellent stability for HER.

## 1 Results and Discussion

Fig. 1 shows the schematic illustration of the fabrication of CoS<sub>2</sub> YS@C spheres. SEM images of CoS<sub>2</sub> YS spheres are shown in Figs. 2(a) and (b), which clearly shows that the diameters of CoS<sub>2</sub> YS spheres are about 2 μm and the surfaces of CoS<sub>2</sub> YS spheres are rough and made up of many small spheres. From the broken CoS<sub>2</sub> YS spheres, the YS structures are clearly seen as shown in Fig. 2(b), and the shell thickness is about 200 nm. This unique hollow and YS structures of CoS<sub>2</sub> will provide large surface areas, and they will be beneficial for the transportations of reactant and resultant and the enhancement of active sites. Then CoS<sub>2</sub> YS spheres coated with carbon thin layers (CoS<sub>2</sub> YS@C spheres) were achieved via hydrothermal treatment of CoS<sub>2</sub> YS spheres in glucose solution for 2 h. SEM image of

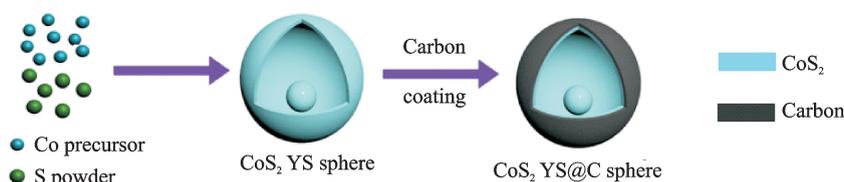


Fig. 1 Schematic of fabrication of CoS<sub>2</sub> YS@C spheres

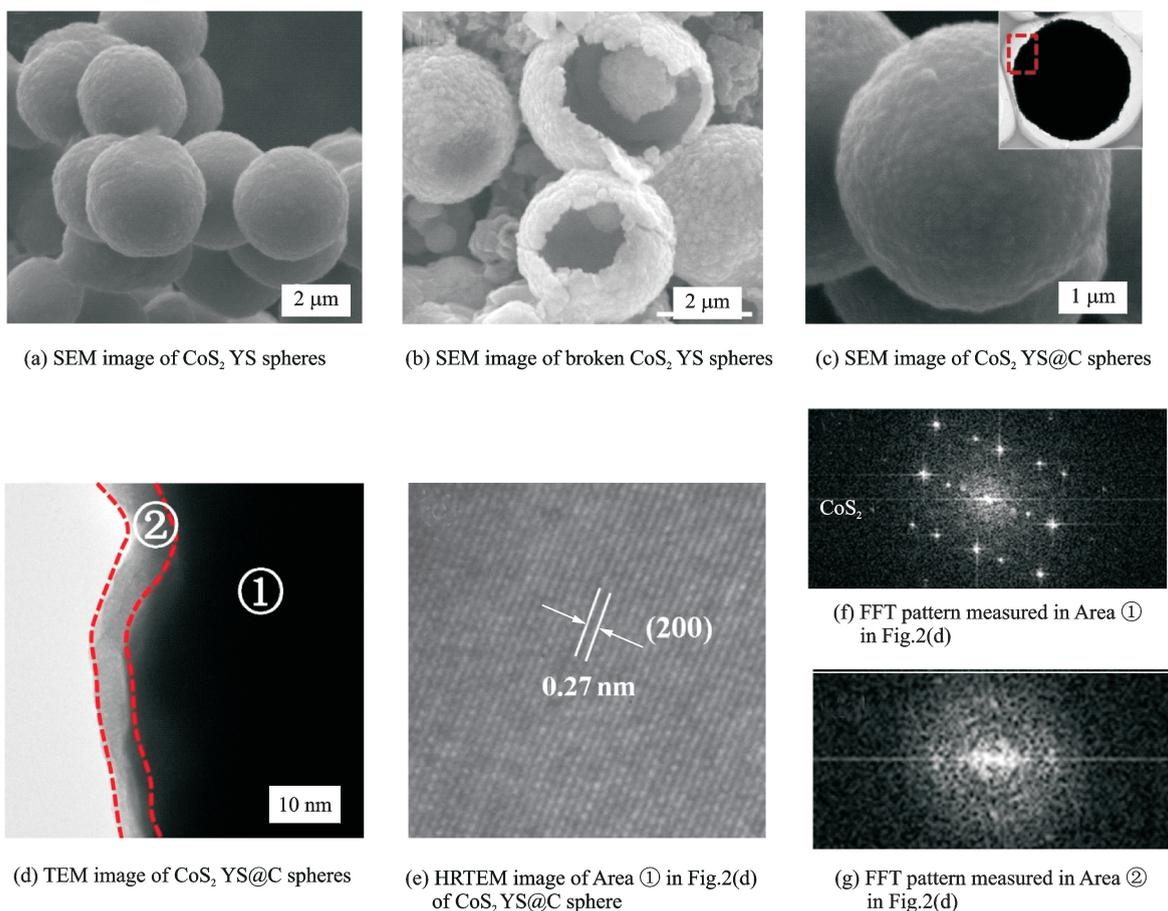


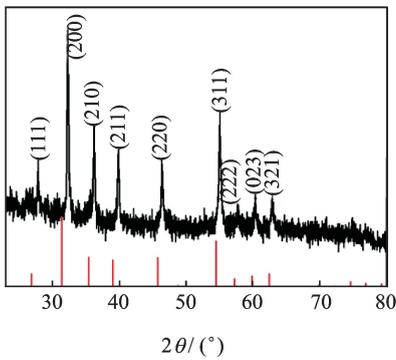
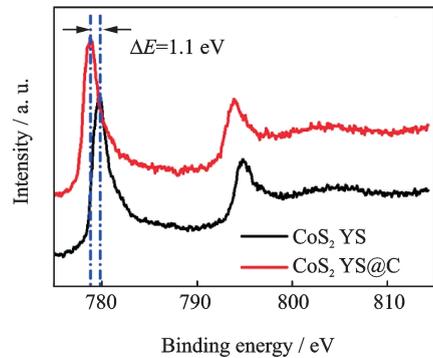
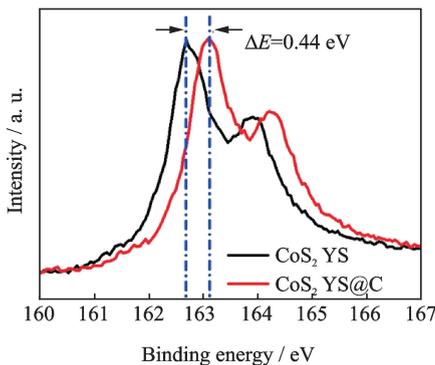
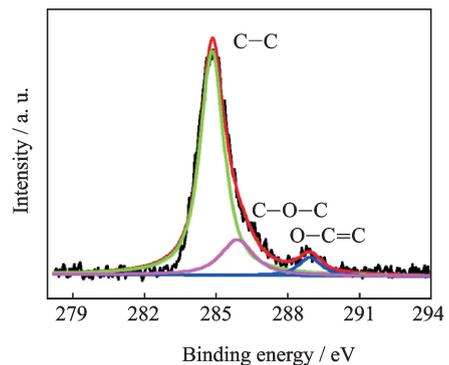
Fig. 2 SEM image of CoS<sub>2</sub> YS spheres, SEM image of broken CoS<sub>2</sub> YS spheres, SEM and TEM images of CoS<sub>2</sub> YS@C spheres, HRTEM image of Area ① in Fig. 2 (d) of CoS<sub>2</sub> YS@C sphere, FFT pattern measured in Area ① in Fig. 2(d), and FFT pattern measured in Area ② in Fig. 2(d)

CoS<sub>2</sub> YS@C spheres in Fig. 2(c) shows carbon layers are uniformly coated on the surfaces of CoS<sub>2</sub> YS spheres. Compared with those of CoS<sub>2</sub> YS spheres, the surfaces of CoS<sub>2</sub> YS@C spheres become smoother because of the coating of carbon layer. To investigate the thickness of carbon layer, HRTEM image of the edge layer of CoS<sub>2</sub> YS@C is measured (Fig. 2(d)), which shows that the carbon layer is uniform with thickness of about 10 nm. Fig. 2(e) shows that the inner CoS<sub>2</sub> owns clear lattice fringes of about 0.226 nm, which corresponds to (200) plane of CoS<sub>2</sub>, and fast Fourier transform (FFT) pattern in Fig. 2(f) indicates the single crystal structure of CoS<sub>2</sub>. The carbon layer is amorphous structure without lattice fringe, and FFT pattern in Fig. 2(g) confirms the amorphous structure of carbon thin layer. XRD pattern of CoS<sub>2</sub> YS@C is shown in Fig. 3(a), which shows that all the diffraction

peaks are attributed to the standard cubic phase of CoS<sub>2</sub> (PDF 65-3322) and no diffraction peak of carbon is seen, furtherly confirming the crystalline structure of CoS<sub>2</sub> and amorphous structure of carbon layer. Here the thickness of carbon layer of CoS<sub>2</sub> YS@C sphere can be well controlled. When the hydrothermal treatment time of CoS<sub>2</sub> YS spheres in glucose solution is 3 h, the surface of CoS<sub>2</sub> YS@C sphere is also uniform and the carbon layer is about 15 nm. When the hydrothermal treatment time of CoS<sub>2</sub> YS spheres in glucose solution is 1 h, the uniform carbon layer is about 6 nm. In order to investigate the effect of carbon layer on the electronic structure of CoS<sub>2</sub>, XPS measurements of CoS<sub>2</sub> YS spheres and CoS<sub>2</sub> YS@C spheres were performed. In Co 2p region (Fig. 3(b)), the peaks of Co 2p<sub>1/2</sub> and 2p<sub>3/2</sub> of CoS<sub>2</sub> YS@C spheres at 793.9 and 778.8 eV both shift to lower binding energy compared with those

of Co  $2p_{1/2}$  and Co  $2p_{3/2}$  of  $\text{CoS}_2$  YS spheres at 795.0 and 779.9 eV, respectively, and the negative shifts are about 1.1 eV. In S 2p region (Fig. 3(c)), the peaks of S  $2p_{1/2}$  and  $2p_{3/2}$  of  $\text{CoS}_2$  YS@C spheres at 163.14 and 164.24 eV both shift to higher binding energy compared with those of S  $2p_{1/2}$  and S  $2p_{3/2}$  of  $\text{CoS}_2$  YS spheres at 162.70 and 163.80 eV, respectively, and the positive shifts are about 0.44 eV. The negative shifts of Co 2p peaks and positive shifts of S 2p peaks well confirm the change of electronic structure of  $\text{CoS}_2$  because of the strong electronic interaction between  $\text{CoS}_2$  and carbon layers. XPS spectrum of C 1s of carbon layers is shown in Fig. 3(d), and it can be deconvoluted into three peaks at 284.8, 285.9 and 288.9 eV, which correspond to the bonds of C—C, C—O—C and O—C=C, respectively. The existences of C—O and O—C=C bonds in carbon layers will make  $\text{CoS}_2$  YS@C spheres more hydrophilic, which is beneficial for the absorption of  $\text{H}_2\text{O}$  for HER<sup>[38]</sup>. In addition, Raman measurements were used to investigate the effect of carbon layer on the electronic structure of  $\text{CoS}_2$ , and Raman spectra of  $\text{CoS}_2$  and  $\text{CoS}_2$  YS@C

spheres are shown in Fig. 3(c). The peaks at 293.5 and 398.6  $\text{cm}^{-1}$  are observed for  $\text{CoS}_2$ , which correspond to the pure librational mode of dumb-bells ( $E_g$ ) and in-phase stretching vibrations of S atom in the dumb-bells ( $A_g$ ), respectively<sup>[39]</sup>, and they are in agreement with the data of  $\text{CoS}_2$  single crystal<sup>[21,40]</sup>. However, for  $\text{CoS}_2$  YS@C, there are about 6  $\text{cm}^{-1}$  negative shifts of  $E_g$  and  $A_g$  peaks (287.5 and 392.6  $\text{cm}^{-1}$ ) compared with those of  $\text{CoS}_2$  YS spheres, as shown in Fig. 3(e), further confirming the change of electronic structure of  $\text{CoS}_2$  because of the strong electronic interaction between  $\text{CoS}_2$  and carbon layers. To determine the content of C in  $\text{CoS}_2$  YS@C sphere, TGA measurements of  $\text{CoS}_2$  and  $\text{CoS}_2$  YS@C spheres were studied in the air and the results are shown in Fig. 3(f). Compared with  $\text{CoS}_2$ , there is slow decrease among the temperature of 100—400 °C for  $\text{CoS}_2$  YS@C, which corresponds to the lose of carbon in the sample. The compositions of C and  $\text{CoS}_2$  in  $\text{CoS}_2$  YS@C spheres are determined to be about 6% and 94% in weight, respectively.

(a) XRD pattern of  $\text{CoS}_2$  YS@C spheres(b) XPS spectra of Co 2p region of  $\text{CoS}_2$  YS and  $\text{CoS}_2$  YS@C spheres(c) XPS spectra of S 2p region of  $\text{CoS}_2$  YS and  $\text{CoS}_2$  YS@C spheres(d) XPS spectra of C 1s region of  $\text{CoS}_2$  YS@C spheres

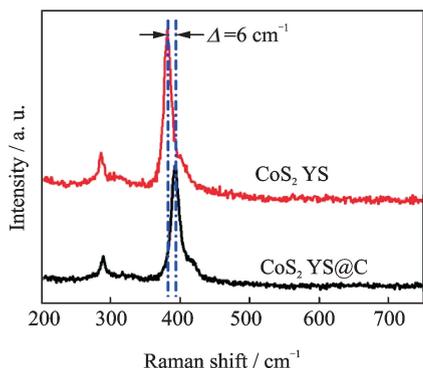
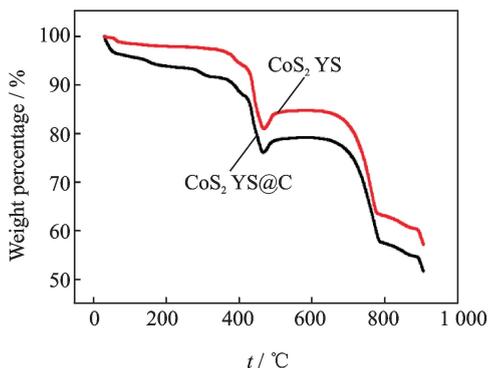
(e) Raman spectra of CoS<sub>2</sub> YS and CoS<sub>2</sub> YS@C spheres(f) TGA curves of CoS<sub>2</sub> YS and CoS<sub>2</sub> YS@C spheres

Fig. 3 XRD pattern of CoS<sub>2</sub> YS@C spheres, XPS spectra of Co 2p region of CoS<sub>2</sub> YS and CoS<sub>2</sub> YS@C spheres, XPS spectra of S 2p region of CoS<sub>2</sub> YS and CoS<sub>2</sub> YS@C spheres, XPS spectra of C 1s region of CoS<sub>2</sub> YS@C spheres, Raman spectra of CoS<sub>2</sub> YS and CoS<sub>2</sub> YS@C spheres, and TGA curves of CoS<sub>2</sub> YS and CoS<sub>2</sub> YS@C spheres

The electrocatalytic activities of CoS<sub>2</sub> YS@C spheres with different carbon layer thickness are studied by linear sweep voltammetry (LSV) in 0.5 M H<sub>2</sub>SO<sub>4</sub> solution at 2 mV/s. When the carbon layer thickness is 10 nm, the electrocatalytic activity of CoS<sub>2</sub> YS@C spheres reaches the highest level (the carbon layer thickness of CoS<sub>2</sub> YS@C spheres was kept to be 10 nm in all the following experiments). The electrocatalytic activities of CoS<sub>2</sub> YS@C spheres, CoS<sub>2</sub> YS spheres and carbon with the same loadings (1.02 mg/cm<sup>2</sup>) were compared and their polarization curves are shown in Fig. 4(a). Obviously, the HER catalytic activity of CoS<sub>2</sub> YS@C is much better than that of CoS<sub>2</sub> YS spheres and the carbon almost has no catalytic activity. The onset potential of CoS<sub>2</sub> YS@C is about 20 mV, which is much lower than those of CoS<sub>2</sub> YS spheres (85 mV), as shown in Fig. 4(b). The overpotential of CoS<sub>2</sub> YS@C at 10 mA/cm<sup>2</sup> is 89.3 mV, which is much smaller than 184.2 mV of CoS<sub>2</sub> YS spheres. In addition, the current density of CoS<sub>2</sub> YS@C at a given potential is much higher than that of CoS<sub>2</sub> YS spheres, as shown in Fig. 4(b). For instance, when the overpotential is 200 mV, the current density of CoS<sub>2</sub> YS@C spheres is about 81.23 mA/cm<sup>2</sup>, which is about 6.5 times higher than that of CoS<sub>2</sub> YS spheres, as shown in Fig. 4(c), suggesting the important role of carbon thin layer for the enhancement of electrocatalytic

activity of CoS<sub>2</sub> YS@C spheres.

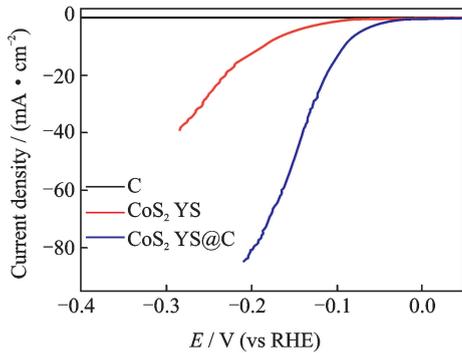
The linear portions of Tafel plots were fit to Tafel equation ( $\eta = a + b \log j$ , where  $j$  is current density,  $b$  is Tafel slope), yielding Tafel slope of about 55 mV/dec for CoS<sub>2</sub> YS@C spheres (Fig. 4(d)), which is much lower than that of CoS<sub>2</sub> (77 mV/dec). The Tafel slope of 55 mV/dec for CoS<sub>2</sub> YS@C indicates Volmer reaction has been taken place<sup>[41-42]</sup>, and the process to convert the protons into absorbed hydrogen atoms on CoS<sub>2</sub> YS@C surfaces becomes rate-determining step during HER. The exchange current density ( $j_0$ ) of the catalyst can be calculated by extrapolating the Tafel plot. As expected,  $j_0$  of CoS<sub>2</sub> YS@C spheres is 0.265 mA/cm<sup>2</sup> (Fig. 5(a)), which is much larger than that of CoS<sub>2</sub> (0.062 mA/cm<sup>2</sup>). Therefore, the CoS<sub>2</sub> YS@C spheres exhibit outstanding HER activity with low onset potential, high current density, low Tafel slope and high exchange current density, which are superior to most of the CoS<sub>2</sub>-based electrocatalysts that have been ever reported in the acidic electrolyte.

In order to further provide the insight to CoS<sub>2</sub> YS@C sphere electrocatalysts, the electrochemical active surface area (ECSA) and electrochemical impedance spectroscopy (EIS) measurements were performed. Though it is difficult to obtain the accurate value of ECSA owing to the unclear capacitive behavior, it can be visualized by double layer capacitance ( $C_{dl}$ ), which is pro-

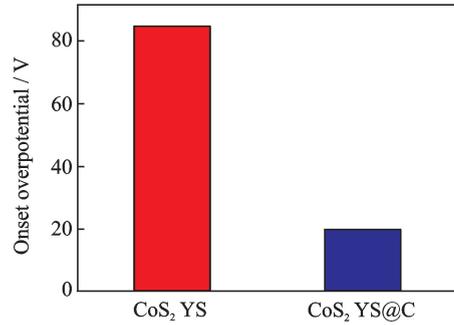
portional to the electrochemical surface area. The calculation of  $C_{dl}$  by cycle voltammograms (CVs) in 0.5 M  $H_2SO_4$  is used to make comparison of ECSA. The  $C_{dl}$  of  $CoS_2$  YS@C spheres is calculated to be  $13.13 \text{ mF/cm}^2$  (Figs. 5(b) and (c)), which is much larger than that of  $CoS_2$  YS spheres ( $6.56 \text{ mF/cm}^2$ ), indicating that  $CoS_2$  YS@C spheres own much larger ECSA than  $CoS_2$  YS spheres. Nyquist plots of  $CoS_2$  and  $CoS_2$  YS@C spheres are shown in Fig. 5(d). The semicircle in the high frequency region is attributed to the charge transfer resistance ( $R_{ct}$ ), which is related to the electrocatalytic kinetics, and a low value of semicircle is consistent with a fast reaction rate<sup>[43]</sup>. From Fig. 5(d), it is clearly seen that  $CoS_2$  YS@C spheres have much lower  $R_{ct}$  than  $CoS_2$  YS spheres, indicating much faster reaction rate for  $CoS_2$  YS@C spheres.

HER electrocatalytic activity of  $CoS_2$  YS@C spheres is significantly better than those of  $CoS_2$

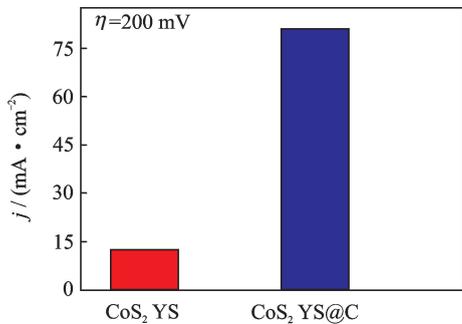
and other  $CoS_2$ -based electrocatalysts reported in the literatures. The enhancement of the catalytic activity of  $CoS_2$  YS@C spheres can be ascribed to the rapid charge transfer based on analyses of EIS results and the large ECSA with more exposed active sites. Actually the better catalytic activity of  $CoS_2$  YS@C spheres as electrocatalysts for HER is also due to the natural properties of  $CoS_2$  YS@C spheres. As we all know, HER activity is strongly correlated with the chemisorption energy of atomic hydrogen to the electrocatalyst surface, and the hydrogen binding energy for an excellent HER electrocatalyst should be neither too high nor too low<sup>[44-45]</sup>. The positive hydrogen binding energy on  $CoS_2$  indicates a weak adsorption of H on  $CoS_2$  surface<sup>[45]</sup>, which will be unfavourable to the reduction of  $H^+$  (i. e., Volmer step). Thus, an optimization of the electronic features is desired. It is notable that the surrounding elements have an important effect on the electron density



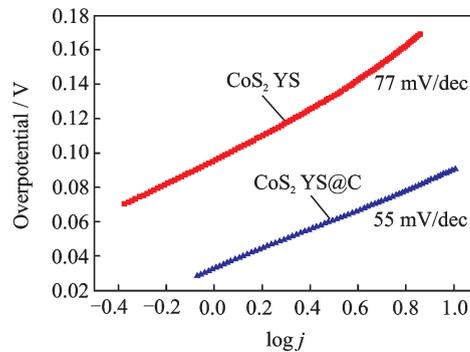
(a) IR-corrected polarization curves of  $CoS_2$  YS spheres,  $CoS_2$  YS@C spheres, and C in 0.5 M  $H_2SO_4$  at 2 mV/s



(b) Comparisons of the onset overpotentials of  $CoS_2$  YS and  $CoS_2$  YS@C spheres



(c) Comparisons of HER current densities of  $CoS_2$  YS and  $CoS_2$  YS@C spheres at the overpotential of 200 mV



(d) Tafel plots of  $CoS_2$  YS and  $CoS_2$  YS@C spheres

Fig. 4 IR-corrected polarization curves of  $CoS_2$  YS spheres,  $CoS_2$  YS@C spheres, and C in 0.5 M  $H_2SO_4$  at 2 mV/s; comparisons of the onset overpotentials of  $CoS_2$  YS and  $CoS_2$  YS@C spheres; comparisons of HER current densities of  $CoS_2$  YS and  $CoS_2$  YS@C spheres at the overpotential of 200 mV; and Tafel plots of  $CoS_2$  YS and  $CoS_2$  YS@C spheres

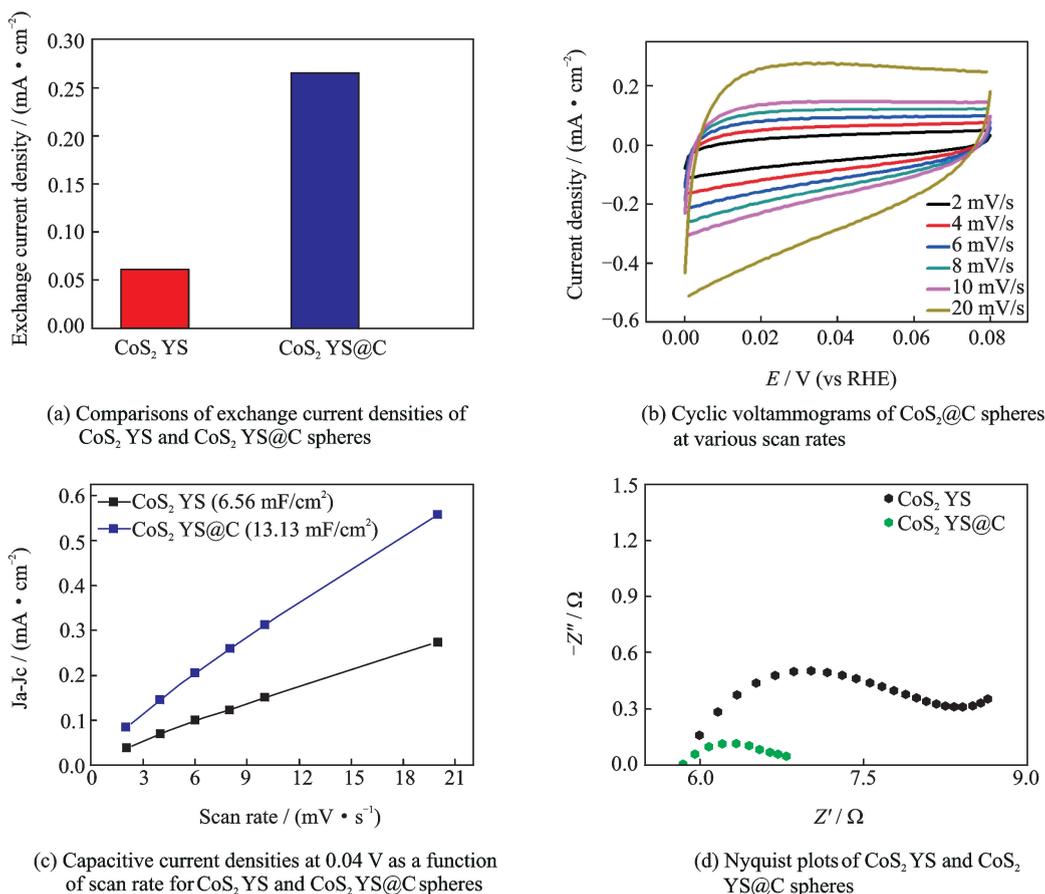


Fig. 5 Comparisons of exchange current densities of CoS<sub>2</sub> YS and CoS<sub>2</sub> YS@C spheres, cyclic voltammograms of CoS<sub>2</sub>@C spheres at various scan rates, capacitive current densities at 0.04 V as a function of scan rate for CoS<sub>2</sub> YS and CoS<sub>2</sub> YS@C spheres, and nyquist plots of CoS<sub>2</sub> YS and CoS<sub>2</sub> YS@C spheres

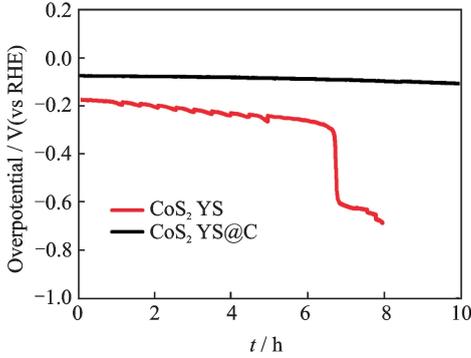
around Co active sites. As we all know, carbon has a lower electronegativity compared with S, so the electron density around Co will increase by embedding carbon onto the surface of CoS<sub>2</sub>. This phenomenon has been well demonstrated by XPS and Raman results. XPS binding energy of Co 2p of CoS<sub>2</sub> YS@C spheres obviously decreases compared with that of CoS<sub>2</sub>, as shown in Fig. 3(b), indicating that the valence of Co in CoS<sub>2</sub> YS@C spheres is below +2 and the electronic density of Co will increase. In addition,  $E_g$  and  $A_g$  peaks of CoS<sub>2</sub> YS@C spheres shifting to low Raman shift compared with that of CoS<sub>2</sub> YS spheres also confirms the strong electronic interactions between CoS<sub>2</sub> and carbon thin layer, as shown in Fig. 3(e). Therefore, the increase of electronic density of Co will consequently enhance the strength of hydrogen binding energy to promote  $H_{ads}$  adsorption and thus will improve HER cata-

lytic activity of CoS<sub>2</sub> YS@C spheres.

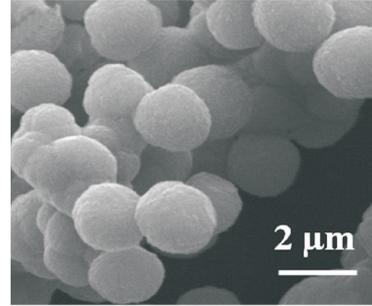
Besides the HER electrocatalytic activity, the stability is also one important criterion in evaluating the performance of electrocatalyst. The long-term HER electrocatalytic stabilities of CoS<sub>2</sub> YS spheres and CoS<sub>2</sub> YS@C spheres were tested through continuous electrolysis at 10 mA/cm<sup>2</sup> in 0.5 M H<sub>2</sub>SO<sub>4</sub> for 10 h. As shown in Fig. 6(a), it is clearly seen that the CoS<sub>2</sub> YS@C spheres exhibit high durability with slight overpotential increase of about 32 mV at 10 mA/cm<sup>2</sup> after 10 h, whereas CoS<sub>2</sub> YS spheres exhibit very poor durability with obvious overpotential increase of about 517 mV after 8 h. To further gain insight into the stability of electrocatalysts, CoS<sub>2</sub> YS spheres and CoS<sub>2</sub> YS@C spheres after stability tests were further studied by SEM and XPS. It is observed that the surface morphology of CoS<sub>2</sub> YS@C still remains very well after stability tests, as shown in

Fig. 6(b). However, for  $\text{CoS}_2$  YS spheres, their surface morphology is seriously damaged after stability tests. So here the carbon thin layer plays a very important role for the protection from structural breakdown of  $\text{CoS}_2$  YS@C spheres, as illustrated in Fig. 6(e). XPS characterization was also performed on  $\text{CoS}_2$  and  $\text{CoS}_2$  YS@C spheres after stability tests. It is clearly seen that the XPS peaks of Co 2p of  $\text{CoS}_2$  YS@C spheres al-

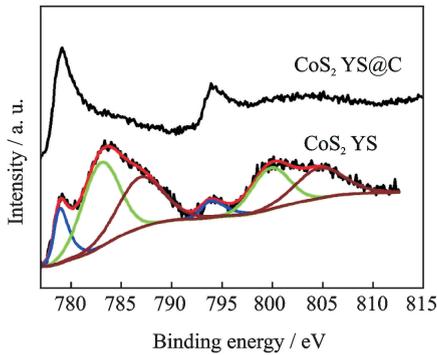
most remain unchangeable compared with those of  $\text{CoS}_2$  YS@C spheres before stability tests, suggesting high chemical stability of  $\text{CoS}_2$  YS@C spheres. However, for  $\text{CoS}_2$  YS spheres, besides the peaks at 778.8 and 794.0 eV of Co 2p, two large new peaks appear at 783.1 and 799.9 eV, as shown in Fig. 6(c), which can be attributed to other forms of Co because of the oxidation of  $\text{CoS}_2$ . In addition, for  $\text{CoS}_2$  YS spheres, a large



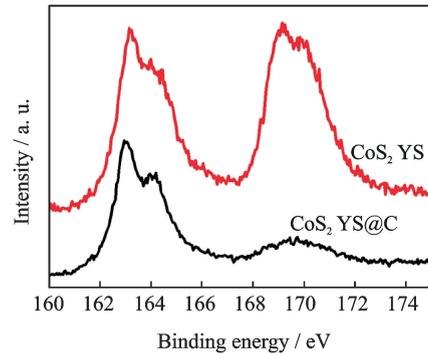
(a) Stability tests of  $\text{CoS}_2$  YS and  $\text{CoS}_2$  YS@C spheres at  $10 \text{ mA/cm}^2$



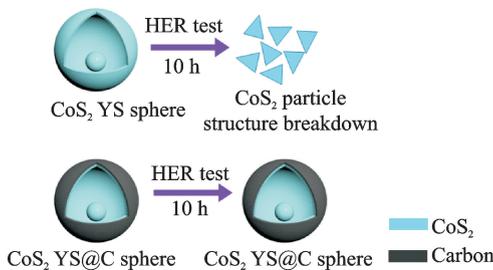
(b) SEM image of  $\text{CoS}_2$  YS@C spheres after stability tests for 10 h



(c) XPS spectra of  $\text{CoS}_2$  YS and  $\text{CoS}_2$  YS@C spheres in Co 2p region after stability tests for 10 h



(d) XPS spectra of  $\text{CoS}_2$  YS and  $\text{CoS}_2$  YS@C spheres in S 2p region after HER tests for 10 h



(e) Schematic of the advantage of  $\text{CoS}_2$  YS@C spheres for long-term HER

Fig. 6 Stability tests of  $\text{CoS}_2$  YS and  $\text{CoS}_2$  YS@C spheres at  $10 \text{ mA/cm}^2$ , SEM image of  $\text{CoS}_2$  YS@C spheres after HER test for 10 h, XPS spectra of  $\text{CoS}_2$  YS and  $\text{CoS}_2$  YS@C spheres in Co 2p region after HER tests for 10 h, XPS spectra of  $\text{CoS}_2$  YS and  $\text{CoS}_2$  YS@C spheres in S 2p region after HER test for 10 h, and schematic of the advantage of  $\text{CoS}_2$  YS@C spheres for long-term HER

peak appears at 169.2 eV in S 2p region, as shown in Fig. 6(d), indicating the evolution of S from CoS<sub>2</sub>. However, for CoS<sub>2</sub> YS@C spheres, the oxidation of CoS<sub>2</sub> or the evolution of S are not observed as shown in Figs. 6(c) and (d). Therefore, after coating carbon thin layers, CoS<sub>2</sub> YS@C spheres can efficiently prevent from the structural breakdown, CoS<sub>2</sub> oxidation or S evolution, which all are beneficial for the improvements of catalytic activity and stability.

## 2 Conclusions

CoS<sub>2</sub> YS spheres coated with carbon thin layers (CoS<sub>2</sub> YS@C spheres) were designed and fabricated as high-performance electrocatalysts for HER in acid media. The unique YS structure provides large space for the reactant and resultant, which is beneficial for the improvement of utilization of active sites. The carbon thin layer coating can efficiently change the electronic structure of CoS<sub>2</sub> for the appropriate hydrogen adsorption energy to promote the Volmer step of HER, and it also can protect CoS<sub>2</sub> from the structural breakdown and can prevent the evolution of S from CoS<sub>2</sub> for the improvement of electrocatalytic activity and stability. Because of the above advantages, CoS<sub>2</sub> YS@C spheres exhibit superior catalytic activity with low onset potential of about 20 mV, small Tafel slope of about 55 mV/dec, and small overpotential of about 90 mV at 10 mA/cm<sup>2</sup>. Especially, CoS<sub>2</sub> YS@C spheres also exhibit high durability with overpotential increase of only about 8% at 10 mA/cm<sup>2</sup> for 10 h. This work provides a new avenue for the design of high-performance electrocatalysts that are unstable during the catalysis process.

## Acknowledgements

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## Supporting Information

# CoS<sub>2</sub> Yolk-Shell Spheres Coated with Carbon Thin Layers as High Active and Stable Electrocatalysts for Hydrogen Evolution Reaction

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### Experimental Sections

#### The synthesis of CoS<sub>2</sub> spheres

CoS<sub>2</sub> spheres were synthesized according to the previous study reported by Lifang Jiao and her collaborators.<sup>1</sup> The details are listed as following: 1. 65 mmol of CoCl<sub>2</sub> · 6H<sub>2</sub>O was dissolved in absolute ethanol and then was transferred into a 40 mL Teflon-lined stainless steel autoclave, then 4.1 mmol of sulfur powder was added into above solution. The Teflon-lined stainless steel autoclave was subsequently stirred for 30 min. The sealed tank was maintained at 240 °C for 24 h. After reaction was over, the autoclave cooled to the room temperature naturally. The precipitations were washed by ethanol three times and by distilled water one time and were collected by centrifugation. Finally, CoS<sub>2</sub> yolk-shell (YS) spheres were obtained after drying at 45 °C for 12 h.

#### The synthesis of CoS<sub>2</sub>@C spheres

The synthesized CoS<sub>2</sub> YS spheres (60 mg) were added into 30 ml 0.05 M glucose solution and then were transferred into a 40 mL Teflon-lined stainless steel autoclave and the sealed tank was maintained at 180 °C for 2 h. After reaction, the autoclave cooled to the room temperature naturally. The precipitations were collected by centrifugation and washed by ethanol and distilled water, respectively. Finally, CoS<sub>2</sub> YS @ C spheres were obtained after drying at 45 °C for 12 h. For comparisons, the different hydrothermal time, such as 1 h and 3 h, was also used for the fabrication of CoS<sub>2</sub>@C with different thickness of carbon layer.

#### Material characterizations

The scanning electron microscope (SEM) and transition electron microscope (TEM) were undertaken on FEI Quanta 400 and FEI Tecnai G2 F30. Energy dispersive spectrum (EDS) mapping was investigated by INCA 300. X-ray powder diffraction (XRD) analysis was performed on Bruker D8 diffractometer using Cu<sub>K</sub>α radiation. X-ray photoelectron spectroscopy (XPS) was processed using an ESCALAB X-ray photo-electron spectrometer. All XPS spectra were corrected using the C 1s line at 284.6 eV, and curve fitting and background subtraction were accomplished. The Fourier transform Raman (FT-Raman) spectrum was achieved on the Nicolet NXR 9650.

#### Electrochemical tests

CoS<sub>2</sub>@C electrocatalysts were loaded on the glassy carbon electrode (GCE, diameter: 5 mm) for testing in 0.5 M H<sub>2</sub>SO<sub>4</sub> using three-electrode system. Saturated calomel electrode (SCE) and graphite rod were used as reference and counter electrodes, respectively. Typically, 10 mg catalysts and 20 μl 5% Nafion solution were dispersed in 0.48 ml ethanol through 30 min ultrasound to form homogeneous ink. For the tests, 10 μl catalyst ink was loaded onto the surface of GCE (loading 1.02 mg · cm<sup>-2</sup>). All the potentials in this paper were referenced to a reversible hydrogen electrode (RHE) by following equation:

$$E(\text{RHE}) = E(\text{SCE}) + 0.241 + 0.059 \text{ pH} \quad (1)$$

Before measurements, the solution was purged with N<sub>2</sub> for 10 min to pip out O<sub>2</sub> dissolved in the solution. HER electrocatalytic activity of CoS<sub>2</sub> YS@C spheres was studied by linear sweep

voltammetry (LSV) at the scan rate of  $2 \text{ mV} \cdot \text{s}^{-1}$ . Double layer capacitance ( $C_{dl}$ ) was measured by CVs using the same working electrode at the potential window of  $0.19\text{--}0.27 \text{ V}$  vs SCE. CVs were obtained at different scan rates of  $2, 4, 6, 8, 10, 20 \text{ mV} \cdot \text{s}^{-1}$ . After plotting charging current density difference ( $\Delta j = j_a - j_c$  at the current density of  $0.23 \text{ V}$ ) vs the scan rates, the slope, which is twice of  $C_{dl}$ , is used to represent ESCA. The chronopotentiometry at  $10 \text{ mA} \cdot \text{cm}^{-2}$  was measured to test the stability of  $\text{CoS}_2 \text{ YS} @ \text{C}$  spheres. The EIS measurements were conducted at overpotential of  $0.24 \text{ V}$  with the frequency ranging from  $100 \text{ kHz}$  to  $0.1 \text{ Hz}$ . For comparisons, bare C,  $\text{CoS}_2$  and the physical mixture of  $\text{CoS}_2 + \text{C}$  (with the same ratio of  $\text{CoS}_2$  and C as that of  $\text{CoS}_2 \text{ YS} @ \text{C}$  spheres) were also mixed with Nafion solution and ethanol to form homogenous ink, respectively, and they were loaded onto the surface of GCE with the same loadings of  $1.02 \text{ mg} \cdot \text{cm}^{-2}$ . The same tests of C,  $\text{CoS}_2$  and  $\text{CoS}_2 + \text{C}$  as those of  $\text{CoS}_2 @ \text{C}$  spheres were also measured.

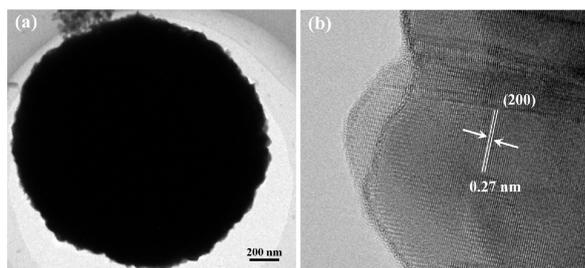


Fig. S1 (a) TEM image of  $\text{CoS}_2 \text{ YS}$  sphere; (d) HR-TEM image of  $\text{CoS}_2$  sphere

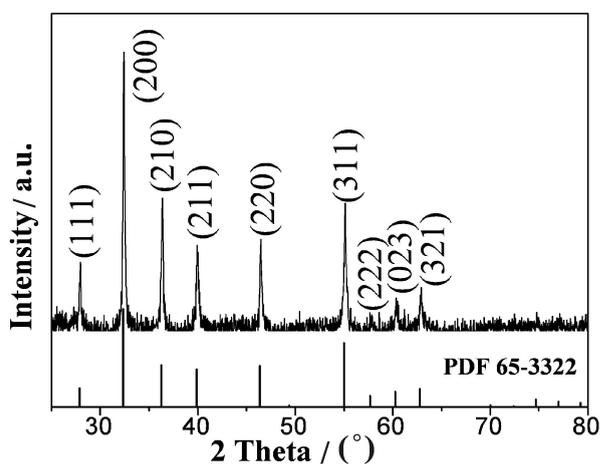


Fig. S2 XRD pattern of  $\text{CoS}_2$  spheres

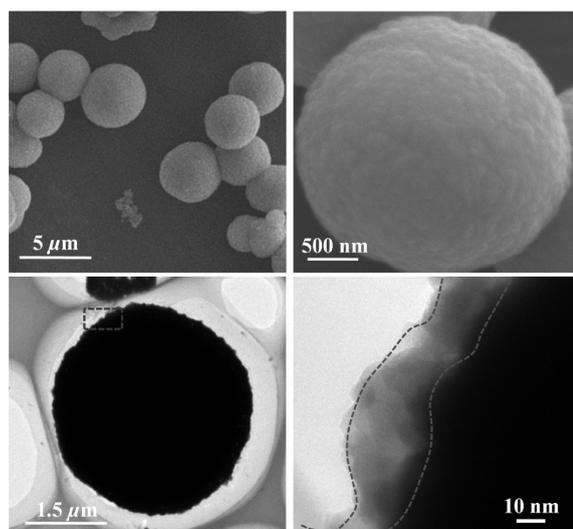


Fig. S3 (a) SEM images of  $\text{CoS}_2 \text{ YS} @ \text{C}$  spheres with different magnifications; (c) TEM image of  $\text{CoS}_2 \text{ YS} @ \text{C}$  spheres and (d) HRTEM image of the wall of  $\text{CoS}_2 @ \text{C}$  sphere

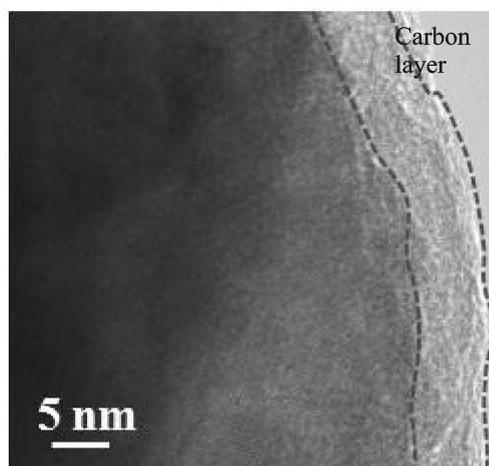


Fig. S4 TEM image of the wall of  $\text{CoS}_2 \text{ YS} @ \text{C}$  sphere with carbon layer thickness of  $\sim 6 \text{ nm}$

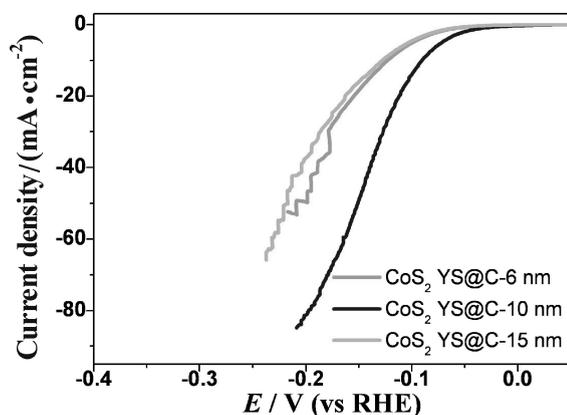


Fig. S5 IR-corrected polarization of  $\text{CoS}_2 \text{ YS} @ \text{C}$ -6 nm,  $\text{CoS}_2 \text{ YS} @ \text{C}$ -10 nm and  $\text{CoS}_2 \text{ YS} @ \text{C}$ -15 nm

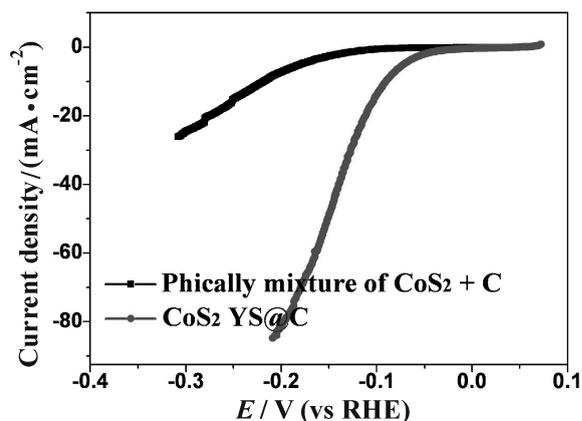


Fig. S6 IR-corrected polarization of CoS<sub>2</sub> YS@C spheres and the physically mixture of CoS<sub>2</sub> + C

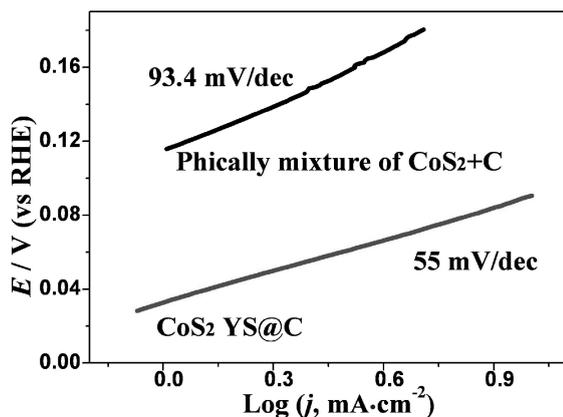


Fig. S7 Tafel curves of CoS<sub>2</sub> YS@C spheres and the physically mixture of CoS<sub>2</sub> + C

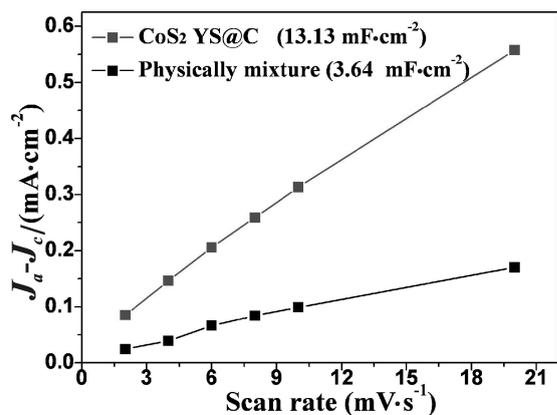


Fig. S8 Capacitive current densities at 0.04 V as a function of scan rate for CoS<sub>2</sub> YS@C and CoS<sub>2</sub> + C

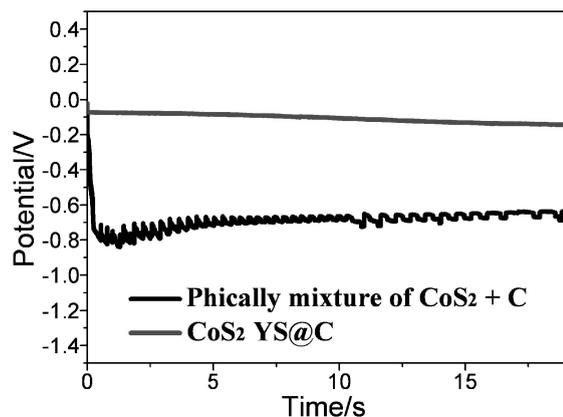


Fig. S9 Stability tests of CoS<sub>2</sub> YS@C and CoS<sub>2</sub> + C at the current density of 10 mA · cm<sup>-2</sup>

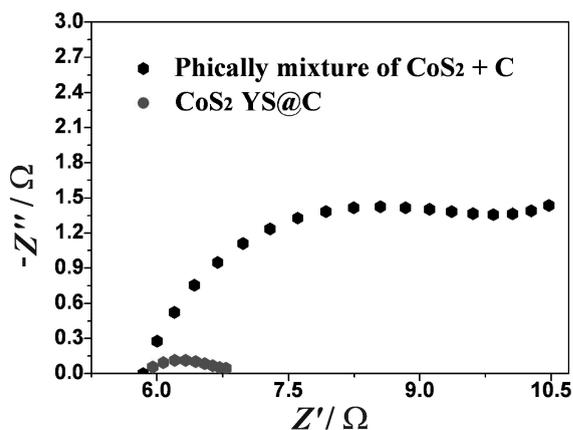


Fig. S10 Nyquist plots of CoS<sub>2</sub> YS@C and the physically mixture of CoS<sub>2</sub> + C

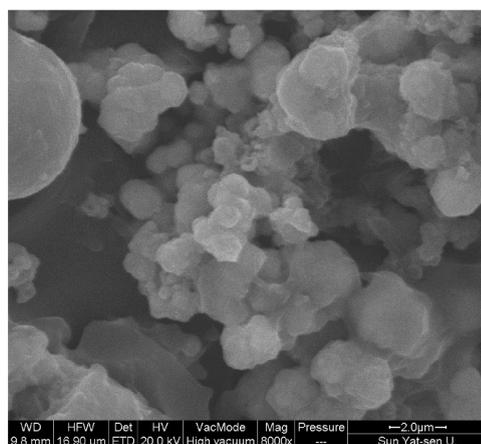


Fig. S11 SEM image of CoS<sub>2</sub> after chronopotentiometry measurement

**Table S1 Comparison of HER electrocatalytic activity of hollow CoS<sub>2</sub> YS@C spheres in acid conditions vis-à-vis other reported CoS<sub>2</sub> or CoS<sub>2</sub>-based HER electrocatalysts**

Electrocatalysts	The overpotentials at 10 mA · cm <sup>-2</sup> /mV	Exchange current densities/(mA · cm <sup>-2</sup> )	Tafel slopes/(mV · dec <sup>-1</sup> )	Loadings/(mg · cm <sup>-2</sup> )	Electrolytes	References
CoS <sub>2</sub> YS@C	90	0.265	55	1.02	0.5 M H <sub>2</sub> SO <sub>4</sub>	This work
CoS <sub>2</sub> films	190	0.00197	51.4	N. A.	0.5 M H <sub>2</sub> SO <sub>4</sub>	2
CoS <sub>2</sub> MWs	158	0.0188	58	25 ± 2	0.5 M H <sub>2</sub> SO <sub>4</sub>	2
CoS <sub>2</sub> NWs	145	0.0151	51.6	1.7 ± 0.3	0.5 M H <sub>2</sub> SO <sub>4</sub>	2
MoS <sub>2</sub> /CoS <sub>2</sub> /CC	87	N. A.	73.4	18.6	0.5 M H <sub>2</sub> SO <sub>4</sub>	3
CoS <sub>2</sub> /CC	288	N. A.	210.7	16.5	0.5 M H <sub>2</sub> SO <sub>4</sub>	3
Co(S <sub>0.73</sub> Se <sub>0.27</sub> ) <sub>2</sub> NWs	104	N. A.	45.3	2.37	0.5 M H <sub>2</sub> SO <sub>4</sub>	4
CoS <sub>2</sub> nanowires		N. A.	47.1	2.24	0.5 M H <sub>2</sub> SO <sub>4</sub>	4
CoS <sub>2</sub> /RGO-CNTs	142	N. A.	51	1.15	0.5 M H <sub>2</sub> SO <sub>4</sub>	5
CoS <sub>2</sub> thin films	192	N. A.	52	N. A.	0.5 M H <sub>2</sub> SO <sub>4</sub>	6
CoS <sub>2</sub> /RGO nanosheets	143	N. A.	285	N. A.	0.5 M H <sub>2</sub> SO <sub>4</sub>	7
CoS <sub>2x</sub> Se <sub>2(1-x)</sub> NWs arrays	129.5	N. A.	44	N. A.	0.5 M H <sub>2</sub> SO <sub>4</sub>	8
CoS <sub>2</sub> NWs	253	0.47	68.7	N. A.	0.5 M H <sub>2</sub> SO <sub>4</sub>	8
P-doped CoS <sub>2</sub> nanosheets	67	0.20	50	2.5 ± 0.1	0.5 M H <sub>2</sub> SO <sub>4</sub>	9
CoS <sub>2</sub> nanosheets	98	N. A.	58	2.0 ± 0.1	0.5 M H <sub>2</sub> SO <sub>4</sub>	9

**N. A. represents the unknown data**

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